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(71) Applicant: Advanced Elastomer Systems, L.P. Akron, Ohio 44311-1059 (US)

(72) Inventors: · Medsker, Robert E. Hartville, Ohlo 44632 (US)

- · Zhao, Jiangun
- Cincinnati, Ohlo 45249 (US) · Gilbertson, Gary W.
- Wadsworth, Ohio 44281 (US)
- · Shen, Kuo-Shein Akron, Ohio 44333 (US)
- · Wang, Donald S.T. Akron, Ohio 44321 (US)
- (74) Representative: Weber, Thomas, Dr. Dipi.-Chem. et al Patentanwälte von Kreisler-Seiting-Werner, Bahnhofsvorplatz 1 (Deichmannhaus) 50667 Köln (DE)
- (54)Preferred process for silicon hydride addition and preferred degree of polymerization for silicon hydride for thermoplastic vulcanizates
- (57) An improved process for the preparation of thermoplastic elastomers by hydrosilylation crosslinking, in which a hydrosilylation crosslinking agent with a low degree of polymerization and/or a hydrositylation crosslinking agent diluted in oil is used.

Description

FIELD OF THE INVENTION

5 [0001] This invention relates to thermoplastic elastomer compositions prepared using hydrositylation crosslinking of the elastomer component of the composition. A thermoplastic elastomer is generally defined as a polymer or blend of polymers that can be processed and revoided in the same way as a conventional thermoplastic material, yet has properties and functional performance similar to that of vulcanized rubber at service temperatures High purformance thermoplastic elastomers in which a highly vulcanized rubbery polymer is infimately dispersed in a thermoplastic matrix.
10 are generally frown as thermoplastic vulcanizated.

BACKGROUND OF THE INVENTION

(9022) Polymer blends, which have a combination of both thermoplastic and elastic properties, are generally so dollared by combining a thermoplastic resin with an elastomeric composition in a way such that the elastomer component is intimately and uniformly dispersed as a discrete particulate phase within a continuous phase of the thermoplastic. Early work with vulcanized rubber components is found in U.S. Pat. No. 3,037,954 which discloses both static vulcanization of the rubber, as well as the technique of dynamic vulcanization wherein a vulcanizatiol elastomer is despersed into a molten resinous thermoplastic polymer and the elastomer is cured white continuously mixing and shear-zo ing the blend. The resulting composition is a micro-get dispersion of cured elastomer in an uncured matrix of thermoplastic polymer.

10033 In U.S. Pat. No. Re. 32,028 polymer blends comprising an olefin flarmoplestic resin and an olefin copolymer are described, wherein the rubber is dynamically vulcanized to a state of partial cure. The resulting compositions are reprocessible. U.S. Pat. Nos. 4,130,534 and 4,130,535 further disclose thermoplestic vulcanizates comprising butyl rubber and polyclefin resin, and olefin rubber and polyclefin resin, and olefin rubber and polyclefin resin, respectively. The compositions are prepared by ryamic vulcanization and the rubber component is cured to the extent that it is essentially insoluble in conventional solvents. A range of crosslinking, or curing, agents for the vulcanization of the rubber are described in the early art, including peroxides, suffure, otherolic resins, radiation, and the like.

[0004] U.S. Pat. No. 4,803,244 generally discusses the use of multifunctional organosition compounds in conjunction with a catalyst as an agent for constitring the rubber component of a thermolestic elastomer by hydrosilytation. Hydrosilytation involves the addition of a silicon hydride across a multiple bond, often with a transition metal catalyst. This patent describes a rhodium catalysed hydrosilytation of EPDM nubber in a blend with polypropylene to produce thermoplastic elastomers having a gel content of up to 34% (after correction for the plastic phase). This degree of vulcarization was achieved only with a high level of catalyst.

55 0005] A turther modification of hydroellylation crosslinking of the rubber in a thermoplestic elastomer composition is disclosed in European Patent Applications Nos. 0776,937-A2 and 651,009. A compatibilizing agent, ordinating in their same molecule a component having an affirity for the trumboar and a component having an affiring for the trumboar and a component having an affiring for the themposition resident in content proposition and is said to improve adhesion between the rubber and resin in order to prevent acolomoration.

SHAMARY OF THE INVENTION

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1006] The present invention is based on the discovery that the process for hydroeilylation crosslinking of the rubber in a thermoplastic vulcanizate (TPV) can be improved by either employing a hydroeilylation crosslinking agent of low degree of polymerization (i.e. up to 30) or diluting the hydroeilylation crosslinking agent in oil before adding it to the TPV or combinations thereof. This combination provides rapid crosslinking of the elastomer to a fully vulcanized state, yet requires an unexpectedly low concentration of the hydroeilylation crosslinking agent and catalyst in order to achieve the cure. In the instant invention no compatibilizer is required in order to produce compositions with excellent mechanical properties.

60 [0007] In a further embodiment of the invention, additives, which react with residual silicon hydride functionality in the thermoplastic elastomer, are incorporated into the process. This results in a composition which has further improved long term heat again or characteristics.

[0008] The compositions produced by the improved process have utility as replacements for thermoset rubber compounds in a variety of applications, particularly where molding or extrusion is involved and the combination of thermosphastic and elastomeric properties provides an advantage. Typical uses include molded articles for automobile underhood parts, engineering and construction materials, mechanical rubber goods, industrial parts such as hose, tubing and gaskies, electrical applications and household goods.

BRIEF DESCRIPTION OF DRAWINGS

[0009]

Figure 1 is a plot of the oil swell value of a TPV versus the degree of polymerization of the silicone hydride.

DESCRIPTION OF THE PREFERRED EMBODIMENTS

[0010] Thermoplastic elastomer compositions may generally be propered by blending a thermoplastic resin and a rubber, then melting the thermoplastic component and mixing the melt until the blend is homogeneous. If a composition of vulcanized rubber in a thermoplastic matrix is desired, crosslinking agents (also referred to as curatives or vulcanizing agents) are added to the blend and crosslinking occurs during the mixing. This latter process is described as dynamic vulcanization.

[0011] A wide range of thermoplastic resins and rubbers and/or their mixtures have been used in the preparation of thermoplastic elastomers, including polypropylene, HDPE, LDPE, VDPE, LLDPE, cyclic oldefin homopolymers or copolymers as well as otelline bodic oppolymers, polystyrene, polypherenies ustified, polypherelyne accellate, activities propylene code and ethylene propylene copolymer (EP) as the thermoplastics, with ethylene propylene dene nubber (EPDM), acryloribite butatiene rubber (NBR), various isobutylene copolymers and/or natural nubber (NBR), as the elastomer. When the elastomer component is crosslined, accent such as Julius, provide, phenologics and lonic compounts are offen used

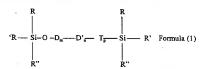
Hydrosilviation Agents

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[0012] Hydrosilylation has also been disclosed as a crosslinking method. In this method a silicon hydride having at least two SIH groups in the molecule is reacted with the achience-achien cautiple bords of an unsaturated (i.e. containing at least two carbon-earbon noticle bonds) public component of the thermoplastic elestomer, in the presence of the thermoplastic resin and a hydrosilylation catalyst. Silicon hydride compounds useful in the process of the invention include methylyhdrogen polysitoxanes, methylflydrogen dimethyl-siloxane copolymers, allryl methyl polysitoxanes, bisc(dimethysily)lalkanes and bisk(dimethysilotparene.

[0013] Preferred silicon hydride compounds may be described by the formula



where each R is independently selected from the group consisting of alkyts comprising 1 to 20 carbon atoms, cycloalkyts comprising 4 to 12 carbon atoms and anyts. In formula (1) it is preferred that each R be independently selected from a group consisting of alkyts comprising 1 to 12 carbon atoms. Even more preferred is R = methyl; R* represents a hydrogen atom, an alkyt or alkoxy group having from 1 to about 24 carbon atoms; and R* represents R or a hydrogen atom. D carposents the group.

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$$\begin{array}{c|c}
R \\
\hline
Si & O \\
R
\end{array}$$

15 D' represents the group

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30 T represents the group



m is an integer having a value ranging from 1 to 50,

- 45 n is an integer having a value ranging from 1 to 50, and
 - p is an integer having a value ranging from 0 to 6. For the purpose of this application the sum of m, n and p will be defined as the degree of polymerization. Preferred degrees of polymerization are up to 30, more preferably from about 20 to 30and most preferably from about 2 or 3 to about 10 or 15 wherein each molecule on average contains at least 2 S+H bonds.
- 50 [0014] Particularly preferred polycogenosilicanes are those in which the silicon atom of the silicon hydride functionality is bound by heteroatoms/atoms having lone pairs of electrons. The preferred polycogenosiloxanes may also be substituted with appropriate functionality permitting solubility in the reaction media. A type of this functionalization is described in U.S. Pat. No. 4,046,930 which teaches alkylation of polycogenosiloxanes. Weight percent of asklyation should not exceed a level that may related on prevent the crossinition reaction rates due to start constraints.
- 150 [0015] It has been discovered that higher gel content in the rubber phase and better physical properties, such as low compression and tension set and low oil swell, can be achieved by diluting the hydrosilylation crosslinking agent in an inert mobile component such as oil prior to addition of it to the components of the thermoplastic vulcanizate. The oil is believed to increase the mobility and diffusion rate of the crosslinking agent in the rubber portion of the thermoplastic

vulcanizate. Desirably, the hydrosilytation crosslinking agent is predicted in oil so that it is from 20 to about 80 weight percent active, more desirably from about 425 to about 75, preferably from about 40 to about 60 weight percent active crosslinking agent with the remainder being substantially oil. As previously set forth, the oil increases the mobility of the hydrositation crosslinking agent (silicon hydride) in the rubber(s) so that it is better dispersed and more effective in centerating crosslinking.

[0016] It has also been discovered that lower molecular weight hydroslylation crosslinking agents are more efficiently dispersed in the components of the TPV and their use results in higher get content and/or more efficient crosslinking of the rubber component. A popular hydrosilylation crosslinking agent has been Dow Corning's DC 2-2822.

structure where m+n is defined as the degree of polymerization and equals about 42 on average and about 30 mole percent of the m+n units are n units. The repeat units with an in subscript can be created by alkylating a repeat unit with an insubscript can be created by alkylating a repeat unit with a subscript m. The n units make the crosslinking agent more soluble in hydrocarbon rubbers. It has been discovered that silicone hydride crosslinkers having a DP of 6 or 8 are more effective than higher DP silicone hydride crosslinkers at equivalent SI-H concentrations. Desirably, the degree of polymerization of the crosslinking agent is up to 30, more desirably from about 2 to about 20 or 30, and preferably from about 2 or 30 about 10 or 15.

[0017] The amount of silicon hydride compound useful in the process of the present invention can generally range from about 0.1 to about 10.0 mole equivalents of SiH per carbon-carbon double bond in the rubber, and preferably is in the range of about 0.5 to about 5.0 mole equivalents of SiH per carbon-carbon double bond in the rubber component of the thermoplastic elastomer.

30 Thermoplastic Resins

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[0018] Thermoplastic resins useful in the compositions produced by the invention include crystalline polyoletin homopolymers and copolymers. They are desirably prepared from monodefin monomers having 2 to 20 cabon abons, asuch as ethylene, propylene, 1-butene, 1-pentene and the like, as well as copolymers derived from linear and cyclic define, with procylene being preferred. As used in the specification and claims the term polypropylene includes homopolymers and expolymers of propylene which can contain about 1 to about 20 wt% of ethylene or an -definic commonmer of 4 to 20 carbon atoms, and mixtures thereof. The polypropylene can be crystalline, is calcactic or syndiciated polypropylene. Commercially available polyolietins may be used in the practice of the invention. Other thermoplastic realins which are substantially inert to the rubber, the silicon hydride and the hydrosilyte-ton catalysts would also be suitable. Blende of themmoplastic resins may also be used.

[0019] The amount of thermoplastic resin found to provide useful compositions is generally from about 5 to about 90 weight percent based on the weight of the rubber and resin (total polymer). Preferably, the thermoplastic resin content will range from about 20 to about 80 percent by weight of the total polymer.

45 Rubbers

[0020] Unsaturated nubbers useful to prepare thermoplastic elastomers according to the invention include monotolefin copolymer rubbers comprising non-poter, rubbery copolymers of two or more monotolefins (EPR rubber), preferably copolymerized with at least one polyene, usually a deen (EPDM rubber). EPDM is a polymer of ethole, por proylene and one or more non-conjugated diene(s), and the monomer components may be polymerized using Ziegler-Natta or metallocene catalyzed reactions, among others. Typically an EPDM rubber has from about 0.5 to about 6 or 10 weight percent of a diene (based on the weight of the polymer) and has a molar ratio of repeat units from ethylene to proylene of from 25.75 to 75.25. Satisfactory non-onlygated dienes include 5-ethylidene-2-norbornene (ENB); 1.46-hexadlene; (HD); 5-methylene-2-norbornene (MNB); 1.6-ctadiene; 5-methyl-1-4-hexadlene; 3-forientfly-1-6-ctadiene; 1.3-cyclopertaclene; 1,4-cyclohexadlene; dicyclopertaclene (DCPD); 5-vinyl-2-norbornene (VNB) and the like, or a combination thereof.

[0021] In one embodiment of the invention, it has been found that rubber having a structure in which the diene monomer has carbon-carbon multiple bonds which are predominately unencumbered, i.e. bonds which are sterically unitin-

dered such as terminal or pendant double bonds, provide a greatly improved rate of cure in the hydrosilytation curing process of the invention. Included in this embodiment are struct: - is in which the bonds either normally are unenumbered or are easily isomenized to form a sterically unencumbered couble bond, which is then rapidly hydrosilytated, e.g. structures from 1,4-hexadiene or ENB. This improvement is particularly eignificant where a hilly cured rubber components desired. The use of rubber in which the diene component is selected from the group consisting of 5-ethyldere-2-notomens, 5-methyl-1,4-hexadiene, 1,4-hexadiene and 5-vinyl-2-notomene is preferred. A structure from 5-vinyl-2-notomene is particularly preferred as a diene component of such rubber.

[0022] Butyl rubbers are also useful in the compositions of the invention. As used in the specification and claims, the term 'butyl rubber' includes copolymers of an isoolatin and a conjugated monoplefin are divinyl aromatic monomers, and the halogenated derivatives of such ocpolymers and terpolymers. The useful butyl rubber copolymers comprise a major portion of isooletin and a minor amount, usually less than 30 w/%, of a conjugated multioletin. The preferred copolymers comprise about 85-99.5 w/% of a C₂/s isooletin such as isobutylene and about 15-0.5 w/% of a multioletin of 4-14 carbon atoms, such as isoprene, butatiene, dimethyl butatiene, 4-methyl-1.4-pentadiene and piperylene. Commercial butyl rubber, useful in the invention, is a copolymer of isobutylene and minor amounts of isoprene. Other butyl co- and terpolymer rubbers are illustrated by the description in U.S. Pat. No. 4.916,180. Isobutylenead/winylberszene is particularly preferred as an elastomer suitable for hydrosilytation crossilinting, as are the halogeneated derivatives of butyl rubber value as othorobutyl and bromoductyl.

[0023] Copolymer and terpolymers including isobutylene (an isoolefin) and aromatic divinyl monomers are particularly preferred as an elastomer suitable for hydrosiylation crosslinking due to the greater reactivity towards crosslinking of the residual carbon to carbon double bonds from aromatic divinyl monomers.

[0024] A highly preferred subset of butyl nubbers are those having carbon to carbon double bonds with greater reactivity towards crosslinking reactions than is typically achieved with residual double bonds derived from conjugated dienes. Examples of these double bonds are the residual double bond in aromatic divinyl compounds such as divinylbenzene monomers after coppolymerization with isobutylene.

5 [0025] The copolymer of at least an iscoletin and an aromatic divinyl monomer desirably comprises from about 80 to about 9.8 weight percent repeating units from an iscoletin (e.g. isomonocletin) of 4 to 7 carbon atoms, from about 0 to about 19.8 weight percent repeating units from an aliphatic polyene (e.g. disins) of 4 to 14 carbon atoms and from about 0.2 to about 5 weight percent repeating units from an aromatic divinyl compound other than said aliphatic polyene of the formula.

Formula II

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where the symbol X represents an aromatic nucleus and the symbols R which may be the same or different represent hydrogen atoms or allyl groups of 1 to 5 calona atoms. X, the aromatic nucleus, can be any aromatic group including those represented by the benzene ring, the naphthalene rings, pyridine ring, and mono or poly allyl substituted versions thereof. The preferred isoclerin is isobutylene, the preferred polyene is isoprene, and the preferred aromatic divinyl compound is divinylberanea. U.S. Patent 3,584,080 is incorporated by reference for a more detailed description of the copolymer. Preferably the repeating units from the polyene is from about 0.1 to about 5 weight percent of the copolymer. Preferably the repeating units from aromatic divinyl compound is from about 0.2 to 4 weight percent (5000 to 30,000 ppm) and more preferably from about 0.5 to about 3 weight percent (5000 to 30,000 ppm) of the copolymer. It is noted that a large portion of the unsaturated repeat units from the aromatic divinyl compound are consumed in account of the unsaturated repeat units from the aromatic divinyl compound are consumed in each of the violence of the compound is consumed in the properties of the preferably the preferably the preferably the preferably the preferable of the unsaturated of peat units from the aromatic divinyl compound are consumed in each of the unsaturated of peat units from the aromatic divinyl compound are consumed in the preferable of the preferable o

[0025] A preferred buyl rubber thus comprises a copolymer of at least an isooletin and an aromatic divinyl compound. The repeat units therein with residual unsaturation from an aromatic divinyl compound are desirably present from about 50 to about 2000 ppm (0.005 to about 0.5 wt% based on the total amount of buyl rubber) and more desirably from about 2000 to about 2500 ppm. The repeat units from an isoolefin desirably are from about 50 to 95.5 wt% and repeat units from a multiplicify floolypen) of 4-14 carbon alons can be absent or present in an amount from about 5.5 to 5.5 wt%.

about 15 wf% based on the total amount of repeat units in the butyl rubber copolymer. A percentage of the repeat units from divinythenzene may be present as repeat units without residual unsaturation. As the repeat units from aromatic divinyl compounds (e.g. divinythenzene) are very reactive to crosslanding, it is sometimes desirable to diffuse the copolymers of isobutylene and aromatic divinyl compounds with more conventional butyl rubbers described in the previous peragraph. A commercial example of a copolymer of isobutylene and divinythenzene is Polyser Butyl XI. 10000 exists before Bayer Corporation. Said Polyser Butyl XI. 10000 exists with about 200-00 pur unsaturated repeat units from divinythenzene. Butyl rubbers are also available from Bayer as XI. 68102 having about 1500 ppm unsaturated repeat units from divinythenzene and Mooney viscosity of 59-75 and as XI. 30102 having about 1500 ppm unsaturated repeat units from divinythenzene and a Mooney viscosity of 59-75 and as XI. 30102 having about 600 ppm unsaturated repeat units from divinythenzene and a Mooney viscosity from about 67-83. The products with designations XI. 68102 and XI. 30102 having about 600 ppm unsaturated repeat units from divinythenzene and a Mooney viscosity from about 67-83. The products with designations XI. 68102 and XI. 30102 having about 6100 ppm unsaturated repeat units from divinythenzene and a Mooney viscosity from about 67-83. The products with designations XI. 68102 and XI. 30102 having about 6100 ppm unsaturated repeat units from divinythenzene and a Mooney viscosity from about 67-83. The products with designations XI. 68102 and XI. 30102 having about 6100 ppm unsaturated repeat units from divinythenzene and a Mooney viscosity from about 67-83. The products with designations XI. 68102 and XI. 30102 having about 6100 ppm unsaturated repeat units from divinythenzene and a Mooney viscosity from about 67-84.

[0027] An unsaturated repeat unit from an aromatic divinyl compound will desirably have the structure:

[0028] Where X and R are as previously defined.

[0029] A close examination of this repeat units reveals that the carbon to carbon double bond has little steric hindrance and mobility with respect to the polymer backbone which may facilitate its reaction with hydrosilation crosslinking agents.

[0030] Another preferred rubber is an acrylic or alleacrylic group functionalized (modified by adding the functional group) opportunity or at least isobutylene and paramethylstyrene. By the term acrylic or alleacrylic group, applicant means acrylic or alleacrylic are combinations thereof. The term alleacrylic is intended to express that the acrylic can have an allyl or alleanyl substituent thereon of 1 to 5 carbon atoms, preferably methyl or eithyl. The functionalization reaction inchives halogeneing the ocopolium of isobutylene and paramethyl syteme as est forth in EP publication no. 0 344 oct (preferably by bromination) and then reacting the brominated polymer with R₂R₅C=CR₁COO 'M* such as illustrated below.

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+ R₂R₂C=CR₂COO · M ·

where R₁ and R₂ are H or an alkyl of 1 to 5 carbon atoms and R₃ is H, an alkyl or an alkenyl of 1 to 5 carbon atoms and M* is a counterion such as H*, Na* ,K*, etc. The amount of acrylic or alkacrylic groups per polymer chain can vary depending on the properties desired. Desirably, the number of moles of acrytic or alkacrylic or combinations thereof, if both are present, is from about 0.1 to about 5 mole %, more desirably from about 0.3 to about 1.5 mole % based on the total moles of repeat units.

Desirably, the copolymer of isobutylene and paramethylstyrene comprises repeat units from at least isobutylene and paramethylstyrene. Other copolymenzable monomers can be present in small amounts. The amount of repeating units from isobutylene is desirably from about 80 to about 99 weight percent, more desirably from about 89 to about 98 weight percent and the amount of repeat units from paramethylstyrene is from about 1 to about 20 weight percent and more desirably from about 2 to about 11 weight percent.

While a preferred embodiment is using the acrylic or alkacrylic group, or combinations thereof, functionalized copolymer of isobutylene and paramethylstyrene as the entire rubber component, it is possible to use a blend of said copolymer with the rubbers described herein. Desirably in a blend of the copolymer with other rubbers the copolymer is a majority by weight of the total rubbers in the thermoplastic vulcanizate.

A further rubber suitable in the invention is natural rubber. The main constituent of natural rubber is the linear 40 polymer cis-1,4-polyisoprene. It is normally commercially available in the form of smoked sheets and crepe. Synthetic polyisoprene can also be used, with the particularly preferred synthetic polyisoprene elastomers being those that contain vinyl functionality pendant to the main polymer chain, i.e. 1,2-enchainments.

Polybutadiene is also a suitable elastomer for hydrosilylation curing with polybutadienes that contain vinyl functionality being the most preferred.

[0035] Blends of any of the above rubbers may also be employed, rather than a single plefinic rubber.

In preparing the compositions of the invention, the amount of rubber generally ranges from about 95 to about [0036] 10 weight percent based on the weight of the rubber and thermoplastic resin (total polymer). Preferably, the rubber content will be in the range of from about 80 to about 20 weight percent of total polymer.

50 Hydrosilylation Catalysts

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It has previously been understood that any catalyst, or catalyst precursor capable of generating a catalyst in situ, which will catalyze the hydrositylation reaction with the carbon-carbon bonds of the rubber can be used. Such catalysts have included transition metals of Group VIII such as palladium, modium, platinum and the like, including complexes of these metals. Chloroplatinic acid has been disclosed as a useful catalyst in U.S. Pat. No. 4,803,244 and European Application No. 651,009, which further disclose that the catalyst may be used at concentrations of 5 to 10,000 parts per million by weight and 100 to 200,000 parts per million by weight based on the weight of rubber, respectively. [0038] It has been found in the process of the present invention that significantly lower concentrations of platinum-

containing catalyst can be used, while obtaining improvement in both the speads of the reaction and the efficiency of the croselinking. Concentrations of catalyst in the range of about 0.01 or 0.1 to about 10, 20 or 40 parts per million by weight of nubber, expressed as platinum metal are effective in rapidly and completely curing the rubber in the process of dynamically vulcanizing blands of thermoplastic resin and rubber. These low catalyst concentrations are particularly effective in combination with a diene-containing rubber having carbon-carbon multiple bonds which are predominately sterically unbindered. Catalyst concentrations of about 0.1 to about 4 or 25 parts per million by weight based on the weight of rubber, expressed as platinum metal, are particularly referred.

[0039] Platinum-containing catalysts, which are useful in the process of the invention, are described, for example, in U.S. Patt. No. 4,578.487; U.S. Patt. No. 2,0972; and U.S. Pattent No. 2,823.218 all of which are incorporated herein by this reference. These catalysts include chloroplatinic acid, chloroplatinic acid hexahydrate, complexes of chloroplatinic acid with symdivinyrletramethytdislicosane, dichloro-bist(hiphenythosphine) pilatinum (II), cis-dichloro-bis(ace-toniritile) pilatinum (III), catalymum (III), patalymum chlorode and platinum oxide. Zero-valent platinum acid. Zero-valent platinum acid. Sero-valent platinum condic. Sero-valent platinum (III), sero-valent platinum condic. Sero-valent p

15 [0040] In order for the catalyst to function most efficiently in the dynamic vulcanization environment, it is important that it is inherently thermally stable, or that its activity is inhibited to prevent too rapid a reaction or catalyst decomposition. Appropriate catalyst inhibitors that are suitable to stablize the platinum catalyst at high remperatures include 1.3,5.74etravity-1.3,5.74etramethylcyclotetrasiloxane and its higher analogs such as vinyl cyclic pentamer. However, other ofefins that are stable above 165° C are also useful. These include maleates, furnarates and the cyclic pentamer.
20 It is also particularly preferred in the invention to use a catalyst that remains soluble in the reaction medium.

Additives

[0041] The thermoplastic elastomer may contain conventional additives, which can be introduced into the composition in the thermoplastic resin, the rubber, or in the blend stither before, during or after the hydrosilylation and curing. Examples of such additives are antioxidants, processing aids, reinforcing and nomeinforcing fillers, pigmants, waxes, rubber processing (o), extender oils, antiblocking agents, antistated speries, ultraviolet stabilizers, pistaticates (including esters), bearing agents, larme retardants and other processing aids known to the rubber compounding art. Such additives may comprise from about 0.1 to about 300 percent by weight based on the weight of the rubber and thermoplastic in the final thermoplastic elastomer product. Fillers and extenders, which can be utilized, include conventional inorganics such as calcium carbonate, clays, slica, tac, tilanium dioxide, carbon black and the like. Additives, fillers or other compounds, which may interfere with the hydrosilylation, should be added after curing reaches the desiral event.

Extender Oil

[00421 The rubber processing or extender oils used in thermoplastic elastomers generally are paraffinic, naphthenic or aromatic oils derived from petroleum fractions. The type will be that ordinarily used in conjunction with the specific rubber or rubbers present in the composition, and the quantity may range from zero to several hundred parts per hundred pans by weight rubber. Important to the efficiency of the catalyst is that the oils and other additives contain no or very low concentrations of compounds that are catalyst inhibitors or that interfere with the activity of the catalyst. These compounds include phosphines, amines, sulfides, thicks or other compounds that may be classified as Lewis bases. Lewis bases, or other compounds that have a pair of electrons available for donation, will react with the platinum catalyst, effectively neutralizing its activity. It has been discovered that the presence of such compounds has a surpris-Ingly detrimental impact on hydrositylation curing in the process of dynamic vulcanization of the rubber component of the thermoplastic elastomer compositions. If the concentration of compounds which have the chemical reactivity of Lewis bases, such as compounds containing sulfur or nitrogen, is maintained at or below a level which provides less than about 1000 ppm and 300 ppm of sulfur and nitrogen respectively, then the amount of platinum catalyst required to promote efficient hydrosilytation curing in dynamic vulcanization can be substantially reduced, usually to the range of about 4 ppm or less, without impact on the cure state of the rubber or the tensile properties of the thermoplastic elastomer product. Concentrations of sulfur and nitrogen below about 500 and 200 ppm respectively are more preferred, and concentrations of less than about 30 ppm sulfur and less than about 100 ppm nitrogen are most preferred. It has been discovered that, even at catalyst concentrations as low as 0.25 ppm, full cure of the elastomer can be achieved if the concentration of sulfur and nitrogen is within the most preferred ranges.

[0043] Most paraffinic petroleum oils for the nubber industry are derived from a coule oil distillation stream. A typical erfining history would include some type of devexing to reduce the pour point, a solvent extraction to physically remove aromatic compounds and a hydrotreating process to chemically modify aromatic structures. Both extraction and hydroreating result in a net increase in the total concentration of saturated hydrocarbon structures and a net decrease in the total aromatic, sultru and hitrogen-containing compound concentration. The degree of reduction in concentration of

these compounds in the oil is dependent upon the type and severity of the retining employed, and the nature of the crude oil. White and parafinic oils have been treated more extensively than aromatic and napthenic oils and would contain a smaller concentration of aromatic, suffur and/or nitrogen compounds. It is difficult to elucidate the exact chemical structure of these compounds due to their complexity. The tendency of an oil to interfere with platinum catalyzed hydrosilylation is directly related to the concentration of suffur and nitrogen containing compounds, as well as compounds which contain prosphorus, fit, arsenic, aluminum and iron.

Processing

70 (0044) The rubber component of the thermoplastic elastomer is generally present as small, i.e. micro-size, particles within a continuous thermoplastic resin matrix, although a co-continuous morphology on a phase inversion is also possible depending upon the amount of rubber relative to plastic and the degree of cure of the rubber. The rubber is desirably at least partially crosslinked, and preferably is completely or thly crosslinked, it is preferred that the rubber be crosslinked by the process of dynamic vulcanization. As used in the specification and claims, the term dynamic vulcanization are such as the specification and claims, the term dynamic vulcanization are rubber blended with a thermoplastic resin, wherein the rubber is vulcanized under conditions of shear at a temperature at which the mixture will fillow. The rubber is thus simultaneously crosslinked and dispersed as fine particles within the thermoplastic resin matrix, although as noted above other morphologies may exist. Dynamic vulcanization is effected by mixing the thermoplastic elastomer components at elevated temperatures in conventional mixing equipment such as roll mills. Banbury mixers, Brabender mixers, continuous mixers, mixing extruders and the like. The unique characteristic of dynamically cured compositions is that, notwithstanding the late that the rubber component is partially or frilly cured, the compositions can be processed and reprocessed by conventional plastic processing techniques such as extrusion, injection molding and compression moiding. Scrap or flashing can be salveged and reprocessed.

The terms "fully vulcanized" and "fully cured" or "fully crosslinked" as used in the specification and claims means that the rubber component to be vulcanized has been cured or crosslinked to a state in which the elastomeno properties of the crosslinked rubber are similar to those of the rubber in its conventional vulcanized state, apart from the thermoplastic elastomer composition. The degree of cure can be described in terms of gel content, or conversely, extractable components. Gel content reported as percent gel (based on the weight of crosslinkable rubber) is determined by a procedure which comprises determining the amount of insoluble polymer by soaking the specimen for 48 hours in organic solvent at room temperature, weighing the dried residue and making suitable corrections based upon knowledge of the composition. Thus, corrected initial and final weights are obtained by subtracting from the initial weight the weight of soluble components, other than rubber to be vulcanized, such as extender oils, plasticizers and components of the composition soluble in organic solvent, as well as that rubber component of the product which is not intended to be cured. Any insoluble polyolefins, pigments, fillers, and the like are subtracted from both the initial and final weights. The rubber component can be described as fully cured when less than about 5%, and preferably less than 3%, of the rubber which is capable of being cured by hydrosilylation is extractable from the thermoplastic elastomer product by a solvent for that rubber. Alternatively the degree of cure may be expressed in terms of crosslink density. All of these descriptions are well known in the art, for example in U.S. Pat. Nos. 4,593,062, 5,100,947 and 5,157,081, all of which are fully incorporated herein by this reference.

40 [0046] The following general procedure was used in the preparation of thermoplastic elastomers (vulcanizates) by the process of the invention, as set forth in the examples. The thermoplastic reain and iol extended rubber were placed in a heated internal mixer with the hydrosilyation agent and hydrosilyteation catalyst. The hydrosilyteation agent and catalyst can be incorporated into the composition by any suitable technique, for example by injection as solutions in oil or as neat components, although a dilute hydrosilytation solution in oil and a dilute catalyst solution are preferred. Addi-45, tives such as antioxidants, ultraviolet stabilizers and fillers may also be added as a slurry in oil. Masterbatches of the components may also be prepared to facilitate the blending process. The mixture was heated to a temperature sufficient to melt the thermoplastic component, and the mixture was masticated, with added processing oil off desired, until a maximum of mixing torque indicated that vulcanization had occurred. Mixing was continued until the desired degree of vulcanization was achieved.

50 [0047] The order of addition of the hydrosilylation agent and hydrosilylation catalyst was found to be important. Maximum catalyst efficiency was obtained when the hydrosilylation agent was added first to the blend, followed by the hydrosilylation catalyst. The mechanical properties of the thermoplestic elastomer products, as well as the degree of cure, were improved when this order of addition was followed.

[0048] The invention will be better understood by reference to the following examples, which serve to illustrate but on timit the present process. In the examples, the following test methods were used to determine the properties of the thermoplastic elastomer products.

Hardness (Shore A/D)	ASTM D 2240
Ultimate tensile strength (UTS - psi)	ASTM D 412
Ultimate elongation (UE - %)	ASTM D 412
Modulus at 100/300% elongation (M1 or M3-psi)	ASTM D412
Tension set (TS - %)	ASTM D 412
Oil swell (OS - %)	ASTM D 471
Heat aging	ASTM D 573

[0049] The rubber component used in the compositions prepared according to the examples is further identified as follows.

Rubber "A"	EPDM - 2.1% ENB; 52% ethylene
Rubber "B"	EPDM - 5% HD; 55% ethylene
Rubber "F"	EPDM - 3% VNB; 55% ethylene
Rubber "G"	EPDM - 5.5% ENB; 60% ethylene
Rubber *K*	EPDM - 1.1% VNB; 64% ethylene
Rubber "L"	EPDM - 0.7% VNB; 62.6 % ethylene
Rubber M	Butyl XL 10,000 from Polysar
Rubber N	Butyl XL 30,102 from Polysar
Polypropylene A	Polypropylene MFI=0.7
Polypropylene B	Homopolypropylene MFI=20
Extender Oil A	Sunpar 150M from Sun Chemical 585 ppm sulfur, 164 ppm nitrogen
Extender Oil B	Sunpar LW 150M from Sun Chemical 19 ppm sulfur, 103 ppm nitrogen
Extender Oil C	White Oil D-200 from Lyondell 1.0 ppm sulfur, 1.0 ppm nitrogen

EXAMPLE 1

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[0050] Compositions were prepared by the method of the invention as generally described above, using polypropylene resian and EPDM rubber comaining various diene components. The plastic and nubber components were melt mixed in a Brabender mixer at 180°C until the polypropylene was melted. Silicone hydride (allylated methyl hydrogen polysioxane) was added dropwise to the melt mix, followed by addition of an oil solution containing latinum [platinate] (ii) hexachitone, dhydrogen reaction product with 2,4,6,8 tetasherhy;2,4,9,6 tetamentyl cyclotralisotizane). The nubber was dynamically vulcanized by mixing the blend until the maximum torque was reached. The product was removed from the mixer, then returned to the mixer and massistated at 180°C for an additional mixer. Plaques were prepared by go compression molding the products of the dynamic vulcanization at 200°C to a thickness of 60 mill and cooling under pressure, and the physical properties were determined using these plaques. All of the products were elastomeric, as defined by ASTIM D1566, i.e. all had tension set values of less than 50%. The compositions and their properties are set forth in Table I.

50061] Example 1 of U.S. Pat. No. 4,803,244 is set forth in comparison to several EPDM rubbers crosslinked with a platinum catalyst. In this comparative example, (Example 1 of U.S. 4,003,244) similar resist and rubber components were dynamically vulcarized by hydrosilyation, but the equivalent of 35 ppm of rhodium metal was used as the catalyst

TABLE

COMPOSITION	'	Patent Ex. 1 of U.S. 4,803,244	"	III	IV
Polypropylene A (parts)	67	50	67	41	41
EPDM Rubber "A" (parts)	100	100	-		
EPDM Rubber "B" (parts)	- 1		100		
EPDM Rubber "F" (parts)		-	-	100	-
EPDM Rubber "G" (parts)			-	-	100
Si-H A (phr)	2.5	6.7	3	2.2	3
Rhodium (ppm)		35	-	-	-
Platinum (ppm)	15		7.6	4	13
	Physica	I Properties			
Hardness (A/D)	93/40	88/26	39 D	69 A	63 A
UTS (psi)	2500	769	2210	1080	905
UE (%)	405	240	330	211	406
M1	1750	305	1510	636	408
TS (%)	22	43	24	-	-
Gel (%) (corrected for plastic phase)	95	15	92	99	90

[0052] It can be seen that the use of much lower levels of platinum catalyst in the hydrosilylation crosslinking of Rubbers A, B, F and G (EPDM rubbers containing 2.1% ENB, 5% HD, 3% VNB, and 5.5% ENB) results in a marked increase in the level of crosslinking (as reflected by gel content) and improved tensite properties in the thermoplastic elastomer, as compared to the use of rhodium as the catalysts.

EXAMPLE 2

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[0053] In order to study the effect of extender oil, compositions were prepared as generally described above using polypropylene reein and EPDM rubber. Mesterbatches of rubber were prepared containing three different extender oils with progressively lower aromatic fractions and lower sulfur and nitrogen concentrations. The mesterbatch composition was 100 parts rubber, 100 parts extender oil, 42 parts clay, 2 parts zinc oxide and 5 parts wax. The polypropylene was melted. Silicone hydrole (3 pin) was added dropwise to the mix, followed by addition of an oil solution containing platinum catalyst at various levels. The rubber was dynamically vulcarized by mixing the blend nuffil maximum torque was reached. Additional processing oil (30 pans) was added after the cure. The product was removed from the mixer, then returned to the mixer and masterated at 180° C for an additional minute. Test specimens were prepared by compression molding the products at 200° C, and the oil swell properties were determined by the test method of ASTM D471, using IRM 903 oil at 125°C for 24 hours. The results, expressed as percent swell (or weight gain) of the specimen, are set torth in Table II.

- 55

TABLE II

	Extend	ler Oil A		Extender	Extender Oil B			Extender Oil C		
Pt Catalyst	os	S:Pt	N:Pt	os	S:Pt	N:Pt	os	S:Pt	N:PI	
0.25 ppm	339	14300	8942	-	-		119	24	55	
0.5	285	7150	4471	105*	227	2808	92.3	12	23	
1.0	222	3575	2236	86.4	114	1404	85.3	6	12	
1.9	140	1882	1177	85.8	60	739	85.6	3.2	6	
2.4	98	1490	932	93.4	48	585	-	T-	-	

^{*}Average of two tests

[10054] The relative degree of swelling in oil is representative of the orosalink density of the vulcanized rubber, i.e., higher crosslink density in the nubber results in lower oil swell values. The data in Table II clearly shows that materials prepared using extender oil which has low amounts of sulfur and nitrogen, and where the mole ratio of sulfur or nitrogen to the platinum in the catalyst is low, results in a thermoplastic elasioner product which is much more effectively crosslinked by hydrosilystich. This effect is seen even at extremely low concentrations of catalyst.

EXAMPLE 3

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25 [0055] The following Table III illustrates the benefit of prediluting the hydrositylation crosslinker in oil.

TABLE III

Effect of Diluting Hydrosilylation Crosslinking Agent								
Composition	1 Cntrl	2 Cntrl	3	4	5	6		
TPV masterbatch (g)*	290	290	290	290	290	290		
Oil added with crosslinker phr		-	1	3	3	9		
Crosslinker phr	3	3	3	3	3	3		
Pt catalyst (ppm**)	6.2	6.6	7.5	6.2	6.2	6.9		
Oil added to masterbatch phr	30	30	29 .	27	27	21		
	Physica	Propertie	es					
Shore A	55	54	54	55	54	55		
TS%	8.5	8.5	8.0	6.0	6.0	7.0		
UTS (psi)	805	770	680	902	800	680		
M100 (psi)	325	310	320	392	350	380		
M300 (psi)	669	620	620	792	720	-		
UE%	444	460	360	378	360	260		
OS%	172	161	159	123	117	119		
Gel%	91.6	89.3	93.5	97.6	96.6	96.0		

^{*(100} parts of rubber) 290g the TPV masterbatch is equivalent to 100g of rubber.
**ppm of Pt catalyst is mg of Pt metal/kg of rubber.

[0056] The TPV masterbatch included measured by weight 100 parts EPDM, 41 parts polypropylene A, 100 parts oil, 42 parts day, 2 parts 27O, and 5 parts wax. A 53.4g portion of this masterbatch was processed in a Brabender mixer using the recipies in Table III. [0057] The EPDM included about 3-3.5 wt % repeating units from ethylidene norbornene and about 0.25 wt % repeating units from vinylidene norbornene.

[0056] The examples 1 - 6 Illustrate that simple addition of concentrated hydrosilylation crosslinker results in about 90% gel while predilution in oil results in about 95% gel and desirable changes in other properties. Compositions where the hydrosilylation crosslinker was added in a 50% or 25% active solution in oil had higher gel content than a composition where it was added as a 75% active solution.

EXAMPLE 4

Table IV below gives the results of oscillating disc rheometer results on hydrosilylation crosslinking of a blend of rubber L and oil with hydrosilylating agents (allicone hydride) of different degrees of polymerization (DP). The rubbler is present as 100 parts by weight and the oil is present at 100 parts by weight. The compositions include 2 prix includes hydrosily and the oil is present at 100 parts by weight. The compositions include 2 prix includes hydrosily and the present as part of 2.75 ppm of pitalinum as catalyst from United Chemical Technology near Philadelphia expressed as plastinum metal per total rubber. The results in this example are associated only with the rubber phase as a plasts present of the present. Compositions prepared using the lower molecular weight silicone hydrides (DP6 and DP9) show more rapid crosslinking and higher sollecular weight silicone hydrides (DP42 and DP54). The compositions for this example were prepared by mixing the rubber and auditives in a Brander mixer at a temperature not exceeding 70°C to prevent premature crosslinking. The oscillating disc rheometer was operated at 200°C.

TABLE IV

The effect of silicone hydride degree of polymerization on reaction kinetics and total torque rise.							
DP of silicone hydride	TS1 (minutes)	TS2 (minutes)	Torque Rise (dN/M)				
6	0.85	3.27	2.55				
8	0.86	3.53	2.53				
42	2.21		1.79				
54	3.18		1.87				

EXAMPLE 5

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[0060] Thermoplastic vulcanizates were prepared from 100 parts EPDM rubber L, 41 parts of polypropylene, 100 parts of oil, 24 parts of oily, and 0.75 ppm of platinum catalyst (expressed as parts by weight platinum metal part sare by weight. The amount of hydrodisylation crosslinking agent was an amount that contained the same number of equivalents of silicon hydride bonds as the DP42 crosslinking parent as 2 parts by weight per 100 parts by weight mitber. These results in Table V indicate that the compositions resulting from the use of DP5 and 8 silicon hydrides have lower weight gain in the Oil Swell Test and also lower compression set. Both of these tests incided a generally higher level of consistinking with equivalent amounts of reactive SH-b honds from the DP6 and 8 silicon hydride materials as compared to that achieved with DP42 and 97 silicon hydride materials. The oil swell values versus degree of polyprimerization of the elicon hydride are also shown in Figure 1.

TABLEV

50	Degree of polym- erization of silicon hydride	Shore A Hardness	UTS (psi)	M100 (psi)	UE%	Oil Swell Weight Gain%	Compression Set
	6	58	792	497	188	84	17
	8	61	934	465	267	98	
55	42	60	808	448	257	106	26
	97	57	700	401	343	155	37

EXAMPLE 6

[0061] Thermoplastic vulcanizates were prepared from 100 parts of EPDM rubber L, 50 parts of polypropylene A, 168 parts of oil, 42 parts of oiley, and 0.5 ppm of platinum catalyst. The parts are parts by weight. The amount and type of hydrosiylation agent varied as shown in Table VI.

[0062] While the physical properties (Shore A, M100, UE, UTS and TS did not vary much in Table VI, the oil swell weight gain, ESR, and AGR Improved with lower degrees of polymerization (OPI) in the hydrosilytation crossilinite SERs is an extrusion surface rating test ASME D 46.1 where lower values indicate a smoother surface. ACR is similar to ASTM D 3835-96 and stands for automatic capillary risonnets. A lower ACR value indicates that a shorter time is 197 required to actrude a specified amount of TPV through a specified orifice diameter. Lower ESR and ACR values for any formulation are desirable. Historically, as one more tightly crosslinked the rubber phase of a TPV, the ESR and ACR properties worsened dramatically feature in Table IV when the amount of 196 or DP42 crosslinker is increased the ESR and ACR values worsen (increase). A comparison between compositions prepared with DP8 crosslinkers and DP42 crosslinkers revealed that the oil swell values are lower for compositions prepared with DP8 crosslinkers. Lower oil swell values in the DP8 crosslinkers have been compositived to the DP8 (properties worself-wed amples) and the TPV. Consequently, one would expect higher ACR and ESR values in DP8 prosslinked sample were about 30% lower than those of the DP42 (2 phr) crosslinked sample were about 30% lower than those of the DP42 (2 phr) crosslinked sample even though the DP8 crosslinked sample were about 30% lower than those of the DP42 (2 phr) crosslinked sample even though the DP8 crosslinked sample lawer oil swell (higher crosslinked sample).

TABLE VI

	DP 42 2.0 phr	DP 42 1.5 phr	DP 42 1.0 phr	DP 8 2.0 phr	DP 8 1.5 phr	DP 8 1.0 phr
Shore A	66	69	70	67	68	70
M100 (ps	346	. 378	354	381	345	331
UE%	593	505	534	540	516	561
UTS (psi	782	741	644	835	713	649
TS%	13	13	15	10.5	13	15
Oil Swell?	% 128	134	152	111	133	138
ESR	88	66	53	61	39	32
ACR	113	105	71	80	76	59

EXAMPLE 7

[0063] This example is a high rubber content TPV comprising 100 parts EPDM rubber L with 5 vinyl-2-notcomene as the diene, 30 parts polypropylene A, 115 parts oil, 10 parts clay, 0.89 ppm of platinum in the form of a platinum catalyst and either 2, 2.5 or 3 phr of a DP24 or DP8 hydrosilylation crosslinker. The physical properties reflect that the DP8 hydrosilylation crosslinker is more efficient on a weight basis than the DP42 material. The Shore A value is generally higher, the MIO is higher, the UE is lower the tension set is the same or lower, and the compossion set is lower. More dramatic improvements are seen in the ESR and ACR. The improvement in ESR indicates that more elastic compounds may be prepared using DP8 crosslinkers without sacrificing surface smoothness. The improvement in ACR indicates that compositions crosslinked with DP8 exhude feater with an equivalent amount of rubber.

TABLE VII

	DP42 2 phr	DP8 2 phr	DP8 2.5 phr	DP8 3 phr
Shore A	58	60	59	58
M100 (psi)	259	280	276	277
UE%	462	426	398	418
UTS (psi)	814	814	773	839

TABLE VII (continued)

	DP42 2 phr	DP8 2 phr	DP8 2.5 phr	DP8 3 phr
TS%	10	10	10	8.75
CompSet%	26.6	23	25	25.2
ESR	113	54	67	67
ACR	490	392	462	457

EXAMPLE 8

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[0064] Copolymers of isobutylene and divinylbenzene (Butyl Rubber M & N) were crosslinked with a hydrosilation crosslinking having a degree of polymerization (DP) of 6 in Table VIII. A control (7) using a hydrosilation crosslinking agent of DP 42 was included for comparison purposes.

[0065] A comparison of compositions 8-11 to composition 7 illustrates that the DP 6 hydrosiation crosslinking agent of compositions 8-11 is much more effective than the DP-42 hydrosilation crosslinking agent of composition 8-11 is much more effective than the DP-42 hydrosilation crosslinking agent and hydrosilation catalyst was gradually reduced in composition 7. In the properties because 20 has a first continuous properties because a first end to the properties because a first end of the properties due to the substitution of butly hubber N for butly nubber M. Butly inubber M is Butly X. 10,000 rubber from Polysar. The catalyst is PC-085 from Dow-Corning diluted to 1 weight percent active in parafficin 0.1 The polyburene oil is Paragol 450 from Exon.

TARLE VIII

			ADLE VIII				
	ORGA	NOSILOXA	NE CURE	BUTYL TI	PV		
Composition	7	8	9	10	11	12	13
Butyl rubber M	50.00g	50.00	50.25	50.51	50.76	34.01	
Butyl rubber N							50.25
Polypropylene B	22.50	22.50	22.61	22.73	22.84	30.61	22.6
Si-H DP= 42	1.50						
Si-H DP=6	i	1.50	1.01	1.01	0.51	0.68	1.0
Catalyst	1.00	1.00	1.01	0.51	0.51	0.68	1.0
(Polybutene) Oil	25.00	25.00	25.13	25.25	25.38	34.01	25.13
Total	100.00	100.00	100.00	100.00	100.00	100.0	100.00
		PHYSICA	L PROPER	TIES			
Hardness A, 5 sec	67	74	74	73	68	87	74
UE %	271	322	252	327	313	249	283
UTS,psi	519	1099	887	1026	668	1172	496
M-100, psi	342	484	486	451	358	830	415
Tear Strength, pli	100	133	116	112	103	184	129
Tension Set,%	20	13	13	15	15	25	36
Comp. Set, 22 hrs, 70C	39	27	28	29	32	43	70

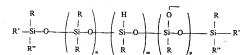
[0066] While the best mode and preferred embodiment of the invention have been set forth in accord with the Patent Statutes, the scope of the invention is not limited thereto, but rather is defined by the attached claims.

Claims

- 1. A process for forming a thermoplastic vulcanizate comprising,
 - a) blending a thermoplastic resin and a rubber in the molten state to form a blend thereof.
 - adding a hydrositylation crosslinking agent diluted in oil to said thermoplastic resin, said rubber, or blends thereof
 - adding a hydrositylation catalyst to said thermoplastic resin, said rubber or blends thereof.
- d) crosslinking said rubber with said hydrosilylation crosslinking agent in said blend.
- A process according to claim 1, wherein said hydrosilylation crosslinking agent is added in the form of an about 20
 to about 80 wt. % blend of hydrosilylation crosslinking agent in an oil.
- A process according to claim 2, wherein said rubber comprises an EPDM copolymer including repeating units from
 5-vinyl-2-norbornene.
 - A process according to claim 2, wherein said hydrosilylation crosslinking agent is added in the form of an about 40
 to about 60 weight percent hydrosilylation crosslinking agent in an oil solution.
- A process according to claim 2, wherein said hydrosilation crosslinking agent has a degree of polymerization from about 2 to about 20.
 - A thermoplastic vulcanizate comprising.
 - a) a thermoplastic resin blended with a crosslinked rubber,
 - b) optionally fillers, extender oils, and plasticizers.
 - c) wherein said crosslinked rubber comprises crosslinks which are the reaction product of a hydrosilylation crosslinking agent having a degree of polymerization of up to 20 reacted with residual unsaturated carbon to
 - carbon double bonds in said rubber in the presence of a hydrositylation catalyst.

 7. A thermoplastic vulcanizate according to claim 6, wherein said nubber comprises an EPDM rubber.
 - A thermoplastic vulcanizate according to claim 6, wherein said hydrositylation crosslinking agent has a degree of polymerization from about 3 to about 15.
 - A thermoplastic vulcanizate according to claim 6, wherein said rubber has residual carbon to carbon double bonds which are sterically unbindered.
 - 10. A thermoplastic vulcanizate according to claim 6, wherein said rubber comprises a copolymer containing at least repeating units from isobutylene and an aromatic divinyl compound.
 - 11. A thermoplastic vulcanizate according to claim 6, wherein said rubber comprises an acrylic or alkacrylic functionalized copolymer from copolymerizing isobutylene and para-methylstyrene.
- 45 12. A process for dynamically vulcanizing a rubber via hydrositylation crosslinking in the presence of a thermoplastic phase, said process comprising,
 - a) mixing a molten thermoplastic resin and a rubber,
 - b) adding to said molten thermoplastic, said rubber, or combinations thereof, a hydrosilylation crosslinking agent and a hydrosilylation catalyst, wherein said hydrosilylation crosslinking agent has a degree of polymerization up to about 20, and
 - c) crosslinking said rubber with said hydrosilylation crosslinking agent.
- A process according to claim 12, wherein said rubber comprises EPDM rubber.
- 14. A process according to claim 12, wherein said rubber has residual carbon to carbon double bond which are sterically unhindered.

- 15. A process according to claim 12, wherein said rubber comprises a copolymer from polymerizing at least isobuty-lene and an aromatic divinyl compound or a copolymer from polymerizing isobutylene and para-methylstyrene which has been functionalized with a pendant acrylic and/or elikacrylic around.
- 16. A process according to claim 12, wherein said hydrositylation crosslinker has a degree of polymerization from about 3 to about 15.
 - 17. A process according to claim 12, wherein said hydrosilylation crosslinker has the formula



wherein each R is independently an allyl of 1 to 20 carbon atoms, a cycloallyl of 4 to 12 carbon atoms or aryl, each 20 R' is independently H or an allyl or alkory group of 1 to 24 carbon atoms, each R' is independently a H or an allyl of 1 to 20 carbon atoms, a cycloallyl of 4 to 12 carbon atoms or an aryl, m is an integer, n is an integer, p is an integer from 0 to 6 and the sum of m and n is an integer from 2 to 20.

18. A process according to claim 17, wherein p is 0.

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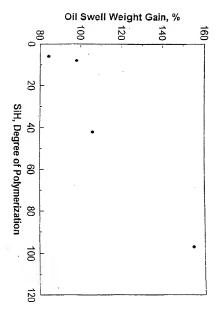


Figure 1



EUROPEAN SEARCH REPORT

EP 99 12 3472

		ERED TO BE RELEVANT		
Category	Citation of document with in of relevant pass	dication, where appropriate,	Relevant to claim	CLARRIPICATION OF THE APPLICATION (INCCL7)
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	The present search report has b	Date of completion of the search		Examine
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ANNEX TO THE EUROPEAN SEARCH REPORT ON EUROPEAN PATENT APPLICATION NO.

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This annex lists the patent family members relating to the patent documents cited in the above—mentioned European search report. The members are as contained in the European Patent Office EDP file on

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